

$\text{Na}_2\text{CrO}_4$ (I) SOLID SOLUTIONS: CRYSTAL CHEMISTRY, DEFECTS AND IONIC  
CONDUCTIVITY

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ABSTRACT

In the system  $\text{Na}_2\text{CrO}_4$ - $\text{CaCrO}_4$  a field of solid solutions  $\text{Na}_{2-x}\text{Ca}_x^{2+}\square_x\text{CrO}_4$  was found extending up to 40%  $\text{CaCrO}_4$ , i.e. 20% cation vacancies ( $\square$ ). These defects are the reason for a remarkable ionic conductivity with a maximum of  $\sigma = 10^{-1} (\Omega\cdot\text{cm})^{-1}$  at 7.5%  $\square$  and at 500°C. In the chromate and the corresponding selenate system four new phases were found:  $\text{Na}_6\text{Ca}(\text{CrO}_4)_4$ ,  $\text{Na}_4\text{Ca}(\text{CrO}_4)_3$ ,  $\text{Na}_4\text{Ca}(\text{SeO}_4)_3$  and high- $\text{CaSeO}_4$ .

METHODS

$\text{CaCrO}_4$  was precipitated from aqueous solutions of  $\text{Na}_2\text{CrO}_4$  and  $\text{CaCl}_2$ . Then mixtures of  $\text{CaCrO}_4$  and  $\text{Na}_2\text{CrO}_4$  were molten for 3 min. in Pt crucibles at 950°C and quenched in air or liquid  $\text{N}_2$  (weight losses 1/2%) or slowly cooled in the furnace (weight loss 2%).

The resulting phases were identified by X-ray powder diffraction (Norelco Diffractometer) or Guinier Camera (AEG) with silicon NBS-SRM 640 as internal standard. High temperature Debye Scherrer patterns were calibrated with  $\alpha\text{-Al}_2\text{O}_3$  ( $a = 4.775$  and  $c = 13.036$  Å at 500°C;  $a = 4.759$  and  $c = 12.992$  Å at 20°C).

For thermal analysis a Du Pont Thermal Analyzer 1090/910 with microcomputer, DSC-cell ( $\leq 700^\circ\text{C}$ ) and 1200°C cell was employed. (For temperature calibration c.f. ref.1). In addition a hot stage (Leitz Company,  $\leq 1350^\circ\text{C}$ ) was used with a polarizing microscope.

Ionic conductivities could be determined by impedance measurements (2) since the  $\text{Na}_2\text{CrO}_4$ (I)ss exhibit no significant electronic conductivity (c.f. ref.3).

THE SYSTEM  $\text{Na}_2\text{CrO}_4$ - $\text{CaCrO}_4$

Various methods were used to determine the Na-rich part of the phase diagram  $\text{Na}_2\text{CrO}_4$ - $\text{CaCrO}_4$ . DTA, DSC and high temperature X-ray (Fig.1) results are included in Fig.2. In addition the phases found by X-ray powder diffraction after quenching or slow cooling were identified. The two polymorphs of  $\text{Na}_2\text{CrO}_4$  are called  $\text{Na}_2\text{CrO}_4$ (III) and  $\text{Na}_2\text{CrO}_4$ (I) in analogy to the isostructural

modification of  $\text{Na}_2\text{SO}_4$ , (III) and (I). The latter has a strongly disordered structure (4).

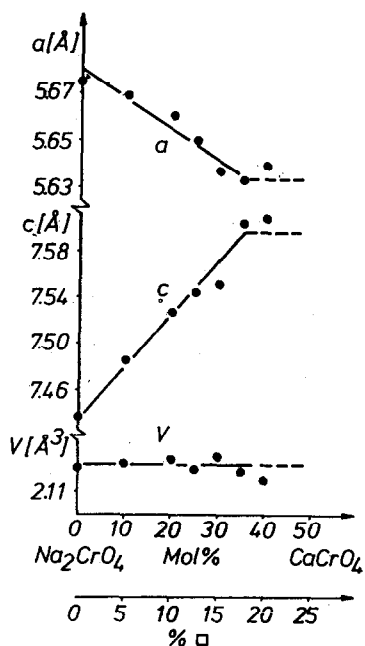


Fig.1 Lattice parameters of  $\text{Na}_2\text{CrO}_4$ (I)ss at  $500^\circ\text{C}$

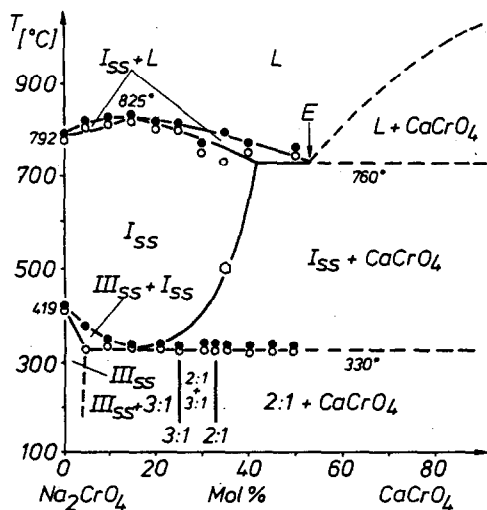


Fig.2  
The system  $\text{Na}_2\text{CrO}_4$ - $\text{CaCrO}_4$   
○ High temperature X-ray results  
● DTA and DSC, heating  
○ DTA and DSC, cooling

The system is characterized by a large field of solid solution of  $\text{Na}_2\text{CrO}_4$  (I) and by two low temperature compounds 2:1 =  $\text{Na}_4\text{Ca}(\text{CrO}_4)_3$  and 3:1 =  $\text{Na}_6\text{Ca}(\text{CrO}_4)_4$ . Their small stability ranges below  $300^\circ\text{C}$  were not determined in detail. During investigations in the corresponding system  $\text{Na}_2\text{SeO}_4$ - $\text{CaSeO}_4$  the isostructural  $\text{Na}_4\text{Ca}(\text{SeO}_4)_3$  and a new modification of  $\text{CaSeO}_4$  with scheelite structure were found (Table 1). The latter was obtained by sintering a pellet at  $400^\circ\text{C}$  for 5 days with subsequent quenching. The 3:1 selenate could not be prepared. In contrast to the corresponding  $\text{Na}_2\text{SO}_4$  (I)ss (ref.4) the  $\text{Na}_2\text{CrO}_4$  (I)ss could not be quenched.

Table 1  
Lattice parameters of new compounds

Complete powder data were sent to JCPDS for publication in the Powder Data file

Compound	Symmetry	a (Å)	c (Å)
$\text{Na}_4\text{Ca}(\text{CrO}_4)_3$	hexagonal	16.315	22.747
$\text{Na}_4\text{Ca}(\text{SeO}_4)_3$	hexagonal	16.348	22.853
$\text{CaSeO}_4$	tetragonal	5.047	11.664

#### IONIC CONDUCTIVITY

The large field of the solid solutions  $(\text{Na}_{2-2x}\text{Ca}_x\text{□}_x)\text{CrO}_4$  implies a massive concentration of cation vacancies □. They provide "hopping paths" for the remaining  $\text{Na}^+$  ions and are the basis of a remarkable ionic conductivity (Fig.3).

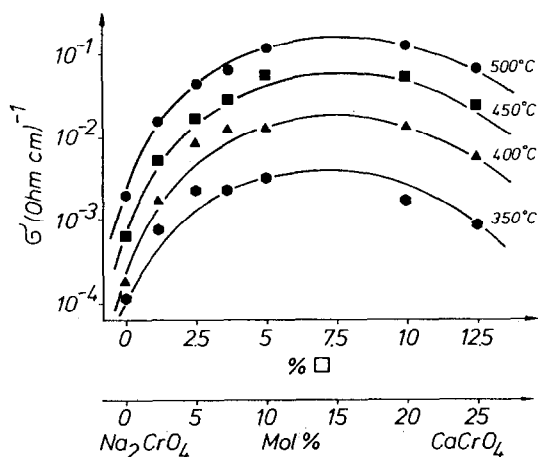


Fig.3 Ionic conductivity of  $\text{Na}_2\text{CrO}_4$  (I)ss

Starting from pure  $\text{Na}_2\text{CrO}_4$ , the conductivity increases strongly and reaches its maximum at about 7.5% vacancies. The following decrease is probably due to defect clustering. The maximum value measured was  $\sigma = 10^{-1} (\Omega \text{ cm})^{-1}$  at  $500^\circ\text{C}$ .

A comparison with the isostructural solid solutions of  $\text{Na}_2\text{SO}_4(\text{I})$  shows a surprising agreement, since also for these materials the maxima of the conductivity curves appear at 7.5% vacancies (3). Due to decomposition reactions corresponding investigations in the system  $\text{Na}_2\text{SeO}_4\text{-CaSeO}_4$  were not possible.

#### ACKNOWLEDGEMENTS

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